

Soft coring: How to get a clarinet out of a flute?

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ABSTRACT

Cutting mozzarella with a dull blade results in poorly shaped slices: the process occurs in a configuration so deformed as to yield unexpectedly curved surfaces. We study the morphogenetics arising from such process through the example of coring: when a thin cylindrical hollow punch is pushed into a soft elastomer, the large transverse expansion occurring during the cutting is responsible for the “clarinet-shape” of the extracted core, which reaches diameters far smaller than those of the tool. With contributions from fracture mechanics and large strain theory, we build a simple yet quantitative understanding of the observed discrepancy, which is shown to occur when the size of the punch is smaller than a characteristic, *tomo-elastic* length scale. Moreover, material nonlinearity and friction appear to play a crucial role in this phenomenon.

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When we use a cookie cutter on a thin layer of dough we obtain the shape we meant, e.g. a gingerbread man! Yet when coring soft elastic materials like elastomers with long cylindrical punches, we are surprised to obtain thin clarinet shapes like those shown in Fig. 1. As a matter of fact, while both ends of the core have the same radius R as the coring blade, the central part of the core reaches a steady-state radius R_{core} which can be significantly smaller – up to a factor 4 – than the former if the cored substrate is thick enough. The overall shape is surprisingly symmetric despite the fact that the coring process has a well defined direction involving large asymmetric strains.

In most cutting operations, we impose the path of the blade (at least we try to). However, as illustrated by our soft coring experiments, the final shape of the cut parts is selected by the large deformation of the sample induced by the blade. In practical situations, in order to enforce a close match between the blade path and the final shape, different strategies are used to minimize this deformation, or to reduce the applied forces. Several lines of work have investigated classical ways to cut efficiently, such as sharpening the blades [1,2], introducing slicing motion [3–6], or minimizing friction [7–9]. Notably, a plethora of cutting tools have been developed historically, specialized for each material, cutting speed, and target chip size [10]. However, when addressing soft materials such as food, gels or living tissues, these strategies can fail and shape control is lost.

The soft coring is investigated here as a robust model experiment for rationalizing the geometrical and material nonlinearities induced by the coupling of large strains and cutting. In this situation, the cutting edge of the blade moves into the material after the latter has been compressed by the applied force required for cutting, resulting into large biaxial extension in a plane perpendicular to the direction of cutting. This is the key geometrical nonlinearity of the problem: the final size of the core results from its elastic recovery once the coring is completed. Furthermore, due to the extreme deformation experienced by the samples, we will see that a quantitative description of the specific material nonlinearity is required to achieve a precise understanding of this size.

A preliminary insight into the scaling of this phenomenon can be obtained by considering the brittle limit, where deformations are small, achieved when using very sharp tools. We consider a cylindrical coring tool of radius R pushed with a force F orthogonally into a soft material with Young's modulus E , as illustrated in Fig. 2. For small strain ($\delta_{el} \simeq 0$), the cut length δ_{cut} can be identified with the vertical blade displacement δ . Invoking Saint-Venant's principle we can expect a steady-state cutting regime when the distance of the cutting edge from the external surfaces is larger than R , resulting in a core of constant final radius $R_{core} < R$. While this provides a rationale for the clarinet shape of the two edges of the core, the focus here will be on the modeling of the ratio $\lambda_R = R/R_{core}$ for steady-state cutting. When initially neglecting friction on the blade and bulk energy dissipation, we can describe the cutting process with fracture mechanics tools

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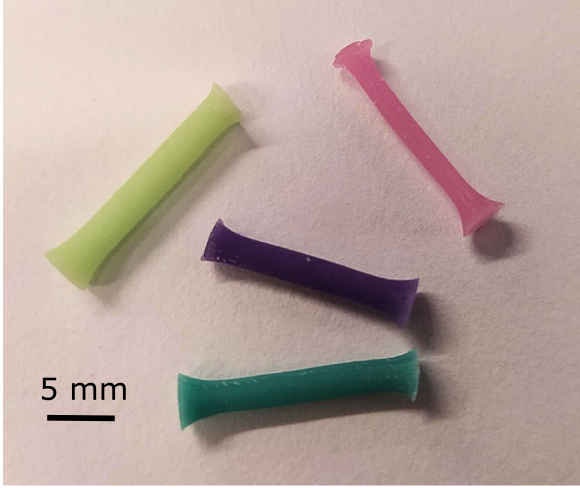


Fig. 1. Some examples of clarinet-shaped cores obtained by cutting elastomers with a cylindrical punch. In these cases, the punch had the same radius of 2 mm, as witnessed by the similar size of the core extremities, but the core radius changes as the samples differ in stiffness.

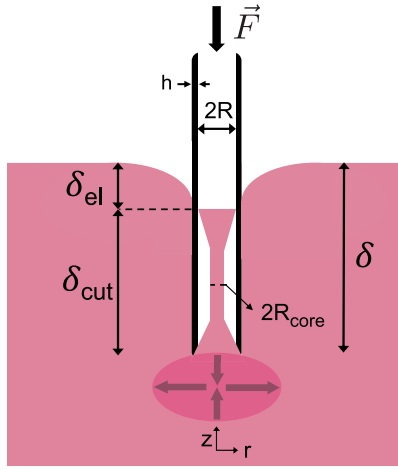


Fig. 2. Sketch of the coring mechanism. The compressed region below the blade of radius R undergoes a contraction of $\lambda_z < 1$ in the vertical direction and an equibiaxial expansion of λ_R in the horizontal plane. While the external portion of the cut sample remains stretched around the blade, the inner part relaxes into the final core shape. The vertical displacement δ of the blade can be split into the cut length δ_{cut} and an elastic component δ_{el} .

such as the Griffith energy release rate. The work performed when the blade is advanced by a distance $d\delta$ is:

$$dW = Fd\delta \simeq Fd\delta_{cut}$$

The average compressive stress at the scale of the blade radius R is $\sigma \sim -F/R^2$. The elastic energy U_{el} stored in the compressed sample can be written as

$$U_{el} \sim \frac{\sigma^2}{E} R^3 \sim \frac{F^2}{ER}$$

which does not depend on the crack length δ_{cut} in the considered steady-state regime. The cutting criterion thus reads

$$G = \frac{dW}{dA} - \frac{\partial U_{el}}{\partial A} = \frac{F}{2\pi R} = \Gamma_{cut} \quad (1)$$

where the cutting energy Γ_{cut} represents the cost for cutting a new unit area of material $dA = 2\pi R d\delta_{cut}$, which results in the

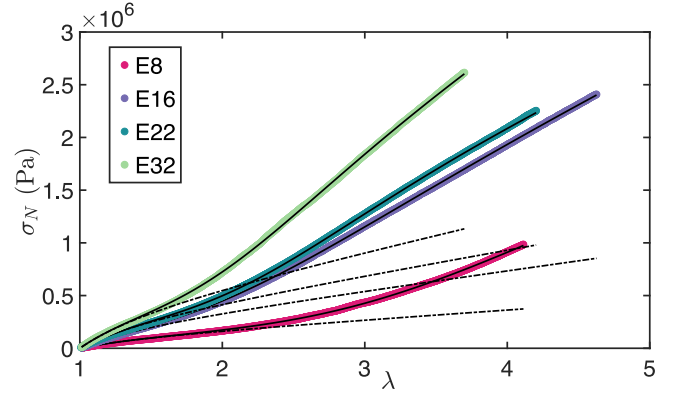


Fig. 3. Uniaxial extension curves (nominal stress against stretch) for the four elastomers (\bullet), and the corresponding fit from the custom model Eq. (11) (continuous black lines), as well as with Neo-Hookean law (dash-dot line). While the Neo-Hookean model only captures the initial softening regime, the custom model allows to describe the behavior of all four materials up to the largest measurable stretch before failure.

cutting force

$$F_{cut} = 2\pi R \Gamma_{cut} \quad (2)$$

The average vertical contraction at the scale R below the blade is thus:

$$\varepsilon_z \sim -\frac{F_{cut}}{R^2 E} \sim -\frac{\Gamma_{cut}}{ER}$$

Due to material incompressibility, this results into a local equibiaxial extension in the horizontal directions $\varepsilon_R = -\varepsilon_z/2$, which sets the scaling for the final radius R_{core} of the cored sample:

$$\varepsilon_R = \frac{R}{R_{core}} - 1 \sim \frac{\Gamma_{cut}}{ER} \quad (3)$$

The outcome of this preliminary scaling analysis is that the shape of the core region is ruled by the ratio between the size R of the cylindrical cutting blade and a characteristic length scale of the material $\ell_{TE} \equiv \Gamma_{cut}/E$, that we will name ‘‘tomo-elastic’’ length scale (from Greek $\tau\omicron\mu\acute{o}\varsigma = cut$). The morphogenetic effect should only be substantial when $R \leq \ell_{TE}$. A rough estimation of ℓ_{TE} can be obtained for elastomers by considering a typical tearing energy of $\Gamma \sim 1000 \text{ J/m}^2$ [7], and modulus $E \sim 1 \text{ MPa}$, which yields millimetric scales.

Following this estimation, we performed systematic experiments by changing the elastic properties of the material and the radius of the coring tools in this regime. As model soft elastic materials, we selected a family of four commercial silicon-based elastomers (Elite Double[®], Zhermack). Their mechanical properties were measured in a standard uniaxial tensile test and reported in Fig. 3. The corresponding elastic moduli are reported in Table 1. One notices that after an initial softening that can be described by a Neo-Hookean hyperelastic model, these materials exhibit strain hardening that is quantitatively captured by a custom equation as will be discussed below.

The samples were cast into cylinders 100 mm in diameter and variable heights up to 30 mm in order to be at least four times larger than the blade radius. The cylindrical blades were made from a set of steel tubes and syringes, with 15 different diameters ranging from 0.5 mm to 8 mm. The blade thickness h remains in approximately constant proportion $R/5$. The tubes were sharpened using a conical milling tool providing variable internal blade angles θ from 10° (sharp) to 90° (flat).

The coring protocol is as follows. A tube is mounted vertically in the jaws of a mandrel attached to the moving part of a testing

Table 1

Material properties of the elastomers. Shore A hardness as provided by the manufacturer. E , a , b , c are the material's parameters obtained by fitting the uniaxial curves in Fig. 3 with Eq. (11). Γ_{cut} and ℓ_{TE} are the values of the cutting energy and tomo-elastic length obtained by fitting the coring data in Fig. 4 with version α of Eq. (8).

Shore A	E (kPa)	a	b	c	Γ_{cut} (J/m ²)	ℓ_{TE} (mm)
8	276	-0.77	0.0064	1.88	1190	4.3
16	559	-0.69	0.079	1.35	1400	2.5
22	709	-0.60	0.046	1.80	1630	2.3
32	938	-0.61	0.082	1.66	1600	1.7

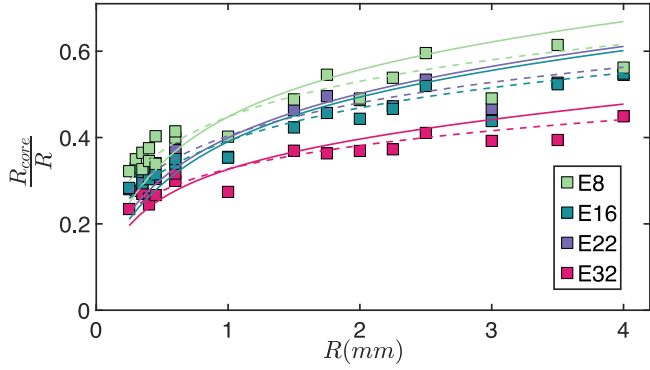


Fig. 4. Measured (\square) coring ratios against the radius of the cylindrical blade. The full and dashed lines represent the predictions of models α and β respectively (cf. Eq. (8)).

machine Instron[®] (model 5965) with a 100 N load cell. An elastomer sample is laid on a flat wooden sheet sitting horizontally underneath the tool. The sample is then cored by moving the tube into the elastomer at a velocity of 10 mm/min until it touches the wooden plate, resulting in a sharp increase in the measured force. The steady-state diameter of the cut part is then measured by taking a high resolution picture with a D800 Nikon camera. The experimental results for the core radius R_{core} and cutting force F_{ss} measured in the steady-state regime are reported in Figs. 4 and 6 as a function of the tool radius R (cf. Fig. 5 for the details on the extraction of F_{ss} from the force-indentation curves). The values of h and θ are not reported here since they did not cause any systematic trend in the data and the amplitude of these effects was small compared to that of both the radius itself and material stiffness.

As predicted by the preliminary linear scaling model, the coring experiments confirm that the samples undergo large strains when the cylindrical blade radius R becomes comparable or smaller than the millimetric scale estimated for the tomo-elastic length ℓ_{TE} . Unsurprisingly, the linear model fails to capture the functional dependency of the core size on the blade radius. Since the observed strains can become larger than 300%, we developed a more advanced model, based on hyperelasticity, in order to account for both finite strains and nonlinear material behavior.

From a geometrical point of view, the elastic indentation δ_{el} becomes larger than the blade radius R , and it can no longer be neglected in front of the length δ_{cut} of the cut material, so that the blade displacement should be written as $\delta = \delta_{el} + \delta_{cut}$ (see Fig. 2). The final core radius R_{core} is selected by the value of the in-plane equibiaxial stretch $\lambda_R \geq 1$ of the elastomer in the compressed region of size R just below the cylindrical blade tip. The compressive force $F_{cut} = -\pi R^2 \sigma_T$ applied by the blade on the elastomer should be related to the local average true stress σ_T by considering the homogeneous uniaxial compression ($\lambda_z \leq 1$) of an incompressible hyperelastic material with energy density

$\mathcal{W}(I_1)$, where I_1 is the first invariant of the Cauchy strain [11,12]:

$$I_1 = \lambda_z^2 + \frac{2}{\lambda_z} \quad \lambda_z = \frac{1}{\lambda_R^2}$$

$$\sigma_N = \frac{d\mathcal{W}(I_1)}{d\lambda_z} = 2 \frac{d\mathcal{W}}{dI_1} \left(\lambda_z - \frac{1}{\lambda_z^2} \right) \quad (4)$$

$$\sigma_T = \lambda_z \sigma_N = 2 \frac{\partial \mathcal{W}}{\partial I_1} \left(\frac{1}{\lambda_R^4} - \lambda_R^2 \right) \quad (5)$$

The value of the local compressive force F_{cut} at the scale R of the blade front during steady-state cutting can also be related to the cutting energy Γ_{cut} by a Griffith-like energy balance similar to Eq. (1). During steady-state cutting all the local parameters such as F_{cut} and λ_R are constant. In the absence of friction the applied force F equals F_{cut} , the elastic component of the displacement δ_{el} is also constant, and the displacement of the loading point is thus simply related to the variation of the cut length δ_{cut} as measured in the reference frame: $d\delta = d\delta_{cut}$. However, the evaluation of the relevant area for the expression of the surface energy term is not obvious under large strain: while the blade front effectively advances along a cylindrical surface of radius R , the surface cut in the unstrained material is a cylinder of smaller radius R_{core} . We chose to initially preserve both choices in the formulation of the model, before discussing their significance and consistency with our measurements:

$$dA = 2\pi \{R; R_{core}\} d\delta_{cut} = 2\pi \frac{R}{\{1; \lambda_R\}} d\delta_{cut}$$

where the term in brackets $\{\alpha; \beta\}$ means the alternative between the two different model hypothesis. The Griffith-like cutting condition then becomes:

$$G = \frac{F_{cut} \{1; \lambda_R\}}{2\pi R} - \frac{\{1; \lambda_R\}}{2\pi R} \frac{dU_{el}}{d\delta_{cut}} = \Gamma_{cut} \quad (6)$$

Although the local strain field has a complex shape, when considering the steady-state cutting regime away from external surfaces, the variation of the total elastic energy U_{el} for a unit cut advance can be estimated by the residual stretch energy in the outer portion of the cut cylinder (the inner part being relaxed as sketched in Fig. 2). For the plane-strain stretch ($\lambda_z = 1$) of a cylindrical cavity of radius R_{core} in an incompressible hyperelastic material, the stretch field is uniquely determined by the stretch λ_R imposed by the punch as [13]:

$$\lambda_r(r) = \frac{1}{\lambda_\theta} = \frac{1}{\sqrt{1 + \frac{R_{core}^2}{r^2} (\lambda_R^2 - 1)}}$$

This allows to express the elastic energy density $\mathcal{W}(I_1)$ as a function of the radial distance r , through the expression of the invariant $I_1(r)$:

$$I_1(r) = \lambda_r^2 + \lambda_\theta^2 + \lambda_z^2 = \frac{1}{1 + \frac{R_{core}^2}{r^2} (\lambda_R^2 - 1)} + \left(1 + \frac{R_{core}^2}{r^2} (\lambda_R^2 - 1) \right) + 1$$

The increment of elastic energy due to a unit depth insertion of the punch is thus:

$$\frac{dU_{el}}{d\delta_{cut}}(\lambda_R) = \int_{R_{core}}^{\infty} \mathcal{W}(r) 2\pi r dr$$

where the finite size of the sample is neglected without loss of generality. According to Eq. (6) the steady-state cutting force F_{cut} can be expressed as:

$$F_{cut} = \frac{\Gamma_{cut} 2\pi R}{\{1; \lambda_R\}} + \frac{dU_{el}}{d\delta_{cut}}(\lambda_R) \quad (7)$$

The value of the radial stretch λ_R can thus be determined as a function of the cutting energy by equating the two expressions for the true stress $\sigma_T = -F_{cut}/\pi R^2$ between Eqs. (5) and (7):

$$\frac{2}{E} \frac{d\mathcal{W}}{dI_1} \left(\frac{1}{\lambda_R^4} - \lambda_R^2 \right) + \frac{1}{\pi R^2 E} \frac{dU_{el}}{d\delta_{cut}}(\lambda_R) + \frac{2\Gamma_{cut}}{ER\{1; \lambda_R\}} = 0 \quad (8)$$

where we chose to divide every term by the elastic modulus E in order to obtain a dimensionless equation for λ_R . This constitutes the main relationship of our finite strain modeling for the shape of the central steady-state portion of the core for an arbitrary hyperelastic potential $\mathcal{W}(I_1)$.

When comparing with the linear model, we remark that the large strain model contains four different sources of non-linearity, which can be attributed either to large geometry changes or to the large strain material response. The large equibiaxial expansion of the compressed region underneath the blade of size R has three different consequences: (1) the hyperelastic uniaxial compression law becomes non-linear due the stretch of the application area (left term); (2) the external part of the cut material remains stretched around the blade and stores elastic energy (middle term); (3) the area of the unstrained cut surface is much smaller than the area swept by the blade (right term). Moreover, the left and middle terms are also affected by the specific nature of the large strain material response encoded in $\mathcal{W}(I_1)$. The competition between these different terms selects a single value of the stretch λ_R as a function of the cutting energy Γ_{cut} , the elastic modulus E and the tool radius R . Since the left and middle terms scale as \mathcal{W} , which scales with E , dimensional analysis confirms that the dimensionless equilibrium stretch λ_R must be written as a function of the dimensionless combination $\Gamma_{cut}/ER = \ell_{TE}/R$.

In order to gain further insight into the relative impact of the four nonlinearities, let us first focus on the geometry by restricting our model to a Neo-Hookean energy density functional:

$$\mathcal{W} = \frac{E}{6}(I_1 - 3)$$

The second term in Eq. (8), representing the radial expansion of the cylindrical cavity, has been explicitly evaluated in [13] as

$$\frac{dU_{el}}{d\delta_{cut}}(\lambda_R) = \pi \frac{E}{3} R_{core}^2 (\lambda_R^2 - 1) \log \lambda_R \quad (9)$$

and thus Eq. (8) can be specialized as:

$$\left(\frac{1}{\lambda_R^4} - \lambda_R^2 \right) + \left(1 - \frac{1}{\lambda_R^2} \right) \log \lambda_R + 6 \frac{\Gamma_{cut}}{ER\{1; \lambda_R\}} = 0 \quad (10)$$

The solution of this simplified model does not correctly predict the dependency of our experimental data for λ_R on the radius of the cutting tool, which can only be captured by taking into consideration the full nonlinear material behavior. However, this Neo-Hookean expression allows to appreciate that the middle term (residual cylindrical stretch energy) becomes very small compared to the left term (uniaxial compression) when the radial stretch λ_R is increased for small values of R . Moreover, it is easily shown that when expanding any elastic energy potential $\mathcal{W}(I_1)$ into a series of power law terms $(I_1 - 3)^n$ in order to capture material strain hardening, the contribution to the left term will always remain $n\lambda_R^2$ times larger than the middle term. We can thus comfortably approximate Eq. (8) by ignoring the middle term when evaluating our model for an arbitrary nonlinear constitutive law.

In particular, in order to capture the non-linear behavior of our elastomers, displayed in Fig. 3, which present an initial softening behavior followed by a linear strain hardening of the $\sigma_N(\lambda)$ curve, we use Eq. (4) with the following custom expression:

$$\frac{d\mathcal{W}}{dI_1} = \frac{E}{6(1+a)} \left(1 + \frac{a}{1+b(I_1-3)^c} \right) \quad (11)$$

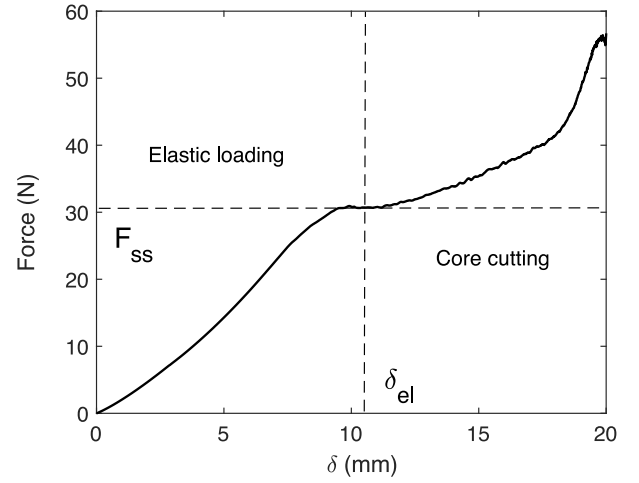


Fig. 5. Typical force–displacement curve obtained during the coring process. Here the elastomer E16 was cast into a 20 mm thick sample, and the punch had a radius of 2 mm. A small drop in force separates the elastic loading regime for small values of the indentation δ from the coring regime for larger indentations, where the force drifts slowly and linearly due to friction on the increasing lateral surface. The final sharp rise is due to the approach of the stiff substrate at the end of the sample. We define as the best estimate for the local steady-state cutting force F_{ss} the measured force at the beginning of the coring process as indicated on the picture.

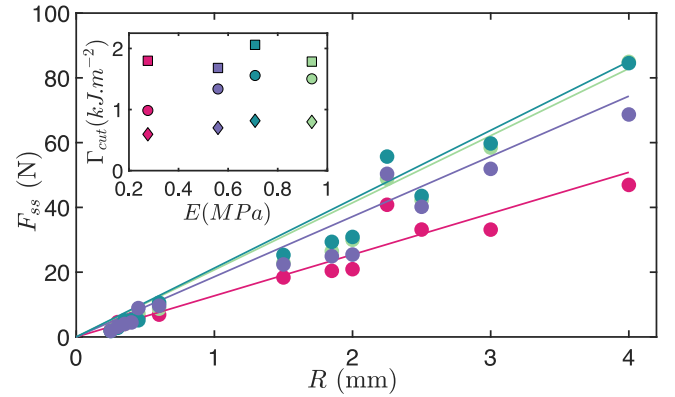


Fig. 6. Scaling of the measured cutting force F_{ss} in the initial stage of steady-state cutting (cf. Fig. 5) against the blade radius R . The inset reports the cutting energy Γ_{cut} estimated as the slope of these force measurements by Eq. (2) (○) and by fitting the core shape data from Fig. 4 with model α (◇) and β (□) defined by Eq. (8).

where E represents the Young modulus, while the three dimensionless parameters a, b, c are phenomenological constants with no specific interpretation, which have the main function of robustly describing the specific non-linear behavior that was measured for each elastomer. The fitted parameters for each elastomer are reported in Table 1.

The corresponding solutions of Eq. (8) (where we dropped the middle term) are plotted in Fig. 4. Both versions α and β of the model can adjust the experimental dataset very well for each elastomer over a wide range of cutting blade radii. However, the values of the tomo-elastic length $\ell_{TE}^{\alpha, \beta} = \Gamma_{cut}^{\alpha, \beta}/E$ that are extracted by the fitting procedure depend on the choice of the model α or β . The corresponding values of $\Gamma_{cut}^{\alpha, \beta}$ are reported in the inset of Fig. 6.

Let us now discuss more extensively the important role of friction in the coring process. When pushing a blade into a soft material we can soundly separate two main contributions of

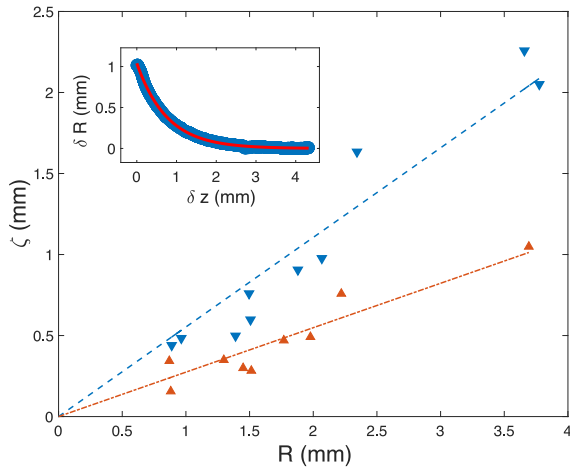


Fig. A.7. Characteristic length ζ associated with the extension of the clarinet bell shape along the core (through Eq. (A.1)) for various values of the coring radius, on the entry side (\blacktriangledown) and exit side (\blacktriangle). Inset: Example exit profile of the clarinet bell shape (\bullet) taken for $R = 3$ mm in elastomer E8. The red line is a fit of Eq. (A.1). (For interpretation of the references to color in this figure legend, the reader is referred to the web version of this article.)

friction, to be respectively associated to the contact of the cut material with the side blade walls and the tip region. The first contribution F_f increases with the progression of the cut length δ_{cut} . In contrast, the second one is localized at the scale of the blade tapered region, and in steady-state cutting conditions it is expected to be invariant during the advance of the blade. We remark that only the latter contributes to the local compressive force F_{cut} , which is responsible for the selection of the lateral stretch λ_R at the moment of cutting, according to the model defined above. This explains why the observed core radius is almost constant during the cutting process, while the total applied force $F = F_{cut} + F_f$ was observed to slowly increase due to increase of the sliding surface (cf. Fig. 5). For that reason we argue that the best observable to be compared with the model parameter F_{cut} is the value of the force F_{ss} measured during the initial stages of steady-state cutting, as defined in Fig. 5.

Following the approach of Irwin [14], the energy dissipated locally due to friction on the tapered region can be added to the fracture energy required to break the polymer network resulting into a global energy cost for cutting a unit area of material Γ_{cut} . However, when considering soft elastomers, the fracture energy term should be evaluated in the unstrained reference (radius R_{core}), while the local friction energy term should be determined by the actual reference (radius R), as described respectively by models β and α introduced above. Although the contributions of fracture and friction to Γ_{cut} were estimated to have the same order of magnitude [1,2,7], the relevance of the friction contribution to Eq. (8) will be dominant for increasing values of λ_R . This explains why only model α provides values of Γ_{cut} which are physically consistent, i.e. lower than the estimates based on the work of the force F_{ss} measured independently during the initial stages of steady-state cutting, as reported in the inset of Fig. 6. This interpretation is also consistent with the observed linear dependency of F_{ss} with the blade radius R (cf. Fig. 6). Future investigations with this soft coring technique might allow independent identification of the contributions of local friction and fracture on the cutting energy Γ_{cut} .

The presented model captures the main features of the steady-state core radius, the selection of which is mainly dominated by the cylindrical blade radius R , the cutting energy Γ_{cut} and

the elastic modulus E , in the form of the ratio between R and the tomo-elastic length $\ell_{TE} = \Gamma_{cut}/E$. The core shape is thus determined by a combination between geometrical and material nonlinearities. Although the physical phenomenon can be qualitatively explained by the geometrical nonlinearity associated to cutting the soft material in a stretched condition, the quantitative relation between the final core radius and the cylindrical blade radius is mainly determined by the details of the nonlinear – strain hardening – material behavior. We remark that if the material were Neo-Hookean, the geometrical effect would be considerably larger than observed in our experiments, since strain hardening has the effect of limiting the stretch in the compressed region. Although friction on the side walls of the blade does not affect the shape of the core, local friction on the blade edge has a dominant effect on the value of the cutting energy Γ_{cut} , and thus on the tomo-elastic length Γ_{cut}/E , which cannot be identified with the elasto-cohesive length Γ_{fract}/E defined for fracture phenomena [15]. As a final remark, friction is also very important for the origin of the two clarinet shapes at the extremities, since it prevents the lateral stretching and thus enforces the core and tool radii to be the same. For the first (incoming) clarinet the lateral stretch is suppressed by static friction of the blade before the initiation of cutting. For the second (outcoming) clarinet the lateral stretch is suppressed by static friction between the sample and the bottom support plate. In both cases, the transient regions shaping the clarinet bells extend over a length of order R from Saint-Venant's principle (see Appendix). Our study opens perspectives into complex programmable surface shaping, in a mundane, accessible setting fitted for an array of design applications.

Declaration of competing interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

Data availability

Data will be made available on request.

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Appendix. Size of the clarinet bell regions

We verify here the claim that the clarinet bell shapes appearing on both ends of our samples indeed develop over regions proportional to the radius of the tool. To do so, we have extracted the profiles of a variety of samples on each side (from the entry point of the tool and from the exit point). The shape of these profiles was then adjusted with an exponential function of the form:

$$\delta R(z) = \delta R_0 \exp\left(-\frac{z}{\zeta}\right) \quad (\text{A.1})$$

where $\delta R(z) \equiv R(z) - R_c$ is the deviation from the core radius at a distance z from the surface of the sample. As seen in the inset of Fig. A.7, the fit is satisfactory and allows to extract the characteristic length scale ζ of the clarinet bell extension. This length scale is displayed in Fig. A.7 as a function of the tool's radius R . While for both the entry and exit we obtain linear relationships, the characteristic length ζ is roughly twice as large for the entry shape than for the exit shape, which might be related to a combination of the difference in boundary conditions

and the different orientation of the out of plane strains with respect to the boundaries.

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